

Development of novel rare earth doped fluoride and oxide scintillators for two-dimensional imaging

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Received 26 August 2011; revised 5 September 2011

Abstract: Two topics were focused. The first one was about the gamma-ray scintillator, Pr³⁺:Lu₃Al₅O₁₂ (LuAG). The second one was about neutron scintillator, Ce³⁺:LiCaAlF₆ and Eu²⁺:LiCaAlF₆ (LiCAF). Those scintillators have been developed very recently for modern imaging applications in the medical and homeland security fields. In both cases, the rare earth ions are playing the crucial role as emission centers. Pr³⁺ in LuAG provided fast 5d→4f transition providing noticeably shorter decay time than that of Ce³⁺. Among several candidate hosts, LuAG showed the best performance. Bulk crystal growth, basic scintillation properties, two-dimensional gamma-ray imaging and positron emission mammography (PEM) application were demonstrated. Due to the international situation, the homeland security was compromised by illicit traffic of explosives, drugs, nuclear materials, etc. and the ways to its improvement became an important R&D topic. For this purpose the Ce and Eu doped LiCAF appeared competitive candidates. Especially, when substitution of ³He neutron detectors was considered, the discrimination ability of gamma-ray from alpha-ray was important. Bulk crystal growth, basic scintillation properties and two-dimensional neutron imaging were demonstrated.

Keywords: Scintillator; gamma-ray detection; neutron detection; oxide; fluoride; Ce, Eu; rare earth

Single crystal scintillator materials are widely used for detection of high-energy photons and particles. There is continuous demand for new scintillator materials with higher performance because of increasing number of medical, industrial, security and other applications^[1].

In the case of modern scintillators, the high light output, good energy resolution, high effective atomic number, fast scintillation response, chemical stability and capability of bulk crystal growth are very important parameters. Especially, medical imaging techniques which are based on the coincidence measurements, requires very fast scintillation decay. Generally, the decay rate Γ of an excited state is given by following equation (Eq. 1)^[2,3]:

$$\Gamma = \frac{1}{\tau} \propto \frac{n}{\lambda_{em}^3} \left(\frac{n^2 + 2}{3} \right)^2 \sum_f | \langle f | \mu | i \rangle |^2 \quad (1)$$

where, Γ is transition probability, τ is decay time, n is refractive index, λ_{em} is emission wavelength, f and i are the ground and excited state wave functions, respectively, and μ is dipole operator.

As far as we are studying the Ce³⁺ doped oxides the emission wavelength is typically from 350 to 550 nm and the decay time determined by the 5d→4f transition of Ce³⁺ is typi-

cally within 20–60 ns. Only rare earth halide host provided the faster Ce³⁺ 5d→4f emission with decay time less than 20 ns^[3–5]. However, their extreme hygroscopicity strongly limits their application potential. Therefore, the emission due to Pr³⁺ 5d→4f transition can be the promising candidate to obtain typically 2–3 times faster scintillation response.

Neutron detection is an essential aspect of interdiction of radiological threats for national security purposes, since plutonium, a material used for nuclear weapons, is a significant source of fission neutrons. The current demand for ³He in commonly deployed neutron detectors and other systems has created an imminent shortage of ³He due to its limited natural resources. Therefore, an alternative technological solution for neutron detection is required in the very near future^[6]. Neutron detector is used not only for security purposes but also in scientific research, e.g. neutron diffraction to investigate the positioning of light elements in the crystal structure. This is also nowadays topic, as the Li based fuel cell becomes highly important. So far, five technologies, boron trifluoride (BF₃) filled proportional detectors^[7,8], boron-lined proportional detectors, scintillating glass fiber detectors, scintillator coated wavelength-shifting fiber detectors, and ⁶Li based or ¹⁰B based bulk neutron scintillators, are proposed as

Foundation item: Project supported by Japan Science and Technology Agency, Regional Research and Development Resources Utilization Program, Ministry of Education, Culture, Sports, Science and Technology of Japanese government, Grant-in-Aid for Young Scientists (A), the joint project between JSPS and AS CR (19686001 (AY)), Czech GAAV Project (M100100910) and the Funding Program for Next Generation World-Leading Researchers, Japan Society for Promotion of Science

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DOI: 10.1016/S1002-0721(10)60621-7

the alternative candidates. In order to realize the simple system for neutron detection, BF_3 and bulk neutron scintillators are the most promising candidates. As BF_3 gas is toxic^[9], it can not be the alternatives for ^3He . Therefore, highly efficient neutron scintillator as well as the bulk crystal growth technologies have to be developed.

Based on the above mentioned background and our recent results, we review two topics: (i) $\text{Pr}^{3+}:\text{Lu}_3\text{Al}_5\text{O}_{12}$ (LuAG) for gamma-ray scintillator^[10-13], (ii) $\text{Ce}^{3+}:\text{LiCaAlF}_6$ ($\text{Ce}:\text{LiCAF}$)^[14,15] and $\text{Eu}^{2+}:\text{LiCaAlF}_6$ ($\text{Eu}:\text{LiCAF}$)^[16] for neutron scintillator. In all the cases, rare earth ions are playing an important role as the emission centers in the above mentioned host structures.

1 Experimental

1.1 Synthesis

1.1.1 Crystal growth of Pr:LuAG Starting materials were Pr_6O_{11} , Lu_2O_3 and $\alpha\text{-Al}_2\text{O}_3$ powders with purity of 99.99%. Iridium crucible was used and heated by radio frequency (RF) induction up to melting temperature (around 2000 °C) under Ar atmosphere. Growth orientation was controlled by the [111]-oriented undoped LuAG single crystalline seed. Crystal rotation rate was ~10 r/min, and the growth pulling rate was 1.0 mm/h.

1.1.2 Crystal growth of Ce doped and Eu doped LiCAF Starting materials were LiF, CaF_2 , AlF_3 , EuF_3 powders with a purity of 99.99%. Graphite crucible was used and heated by radio frequency (RF) induction up to melting temperature (around 900 °C) in $\text{Ar}(95\%)+\text{CF}_4(5\%)$ atmosphere^[17]. Growth orientation was controlled by the [001]-oriented undoped LiCAF single crystalline seed. Crystal rotation rate was ~10 r/min, and the growth pulling rate was 0.5–1.5 mm/h.

1.2 Characterization and performance test

1.2.1 Pr:LuAG After cutting and polishing the as grown crystal, optical properties were studied. Optical transmission spectra were measured using a JASCO V550 spectrophotometer. Luminescence measurements were carried out using spectrofluorometer FLS920 (Edinburgh Instruments) equipped with the hydrogen steady-state, nanosecond and Xe microsecond pulsed flashlamps (IBH Scotland). Spectral range of both equipments is 200 to 900 nm.

For the scintillation properties measurement, the crystals were covered with several layers of Teflon tape and optically coupled to the light entrance window of photo-multiplier tube (PMT) R7600 (Hamatsu) with an optical grease. The high voltage was supplied by ORTEC 556, and signals were read out from the anode of PMT. Then, the signals passed a shaping amplifier ORTEC 570 with 0.5 μs shaping time, converted to digital signals by a multi channel analyzer Pocket MCA 8000A provided by Amptek Co. and recorded in a computer. Decay time was also evaluated using an oscilloscope, Tektronix TDS3034B.

As for the 2 dimensional imaging test, Pr:LuAG pixel array (1 pixel: 2 mm square, 20×20 pixels) were prepared us-

ing BaSO_4 as reflector. BaSO_4 powder was mixed with an adhesive and ethanol. Acrylic-type adhesive which has high transparency around 310 nm was selected. The 0.1 mm thick layer of this mixture was deposited onto each crystal. The pixelated scintillator array was optically coupled with H8500 photomultiplier to study its performance as a gamma camera. The optical coupling was enabled by silicon grease (OKEN 6262A). After coupling, the detector array was irradiated by the 662 keV gamma-ray (^{137}Cs) from the top side.

All the experiments were carried out at room temperature.

1.2.2 Ce:LiCAF and Eu:LiCAF Transmission spectra were obtained with the spectrometer JASCO, V-530. Luminescence measurements were performed with the spectrofluorometer FLS920 (Edinburgh Instruments) equipped with the hydrogen steady-state lamp. ^{241}Am was used as an α -ray source. In order to see the response after thermal neutron excitation, ^{252}Cf neutron radiation was used to excite the sample, Pb plate was inserted to remove gamma-ray. Photoelectron yield was compared with a standard reference ^6Li enriched solid state neutron scintillator, such as Li-glass scintillator (GS20, Saint Gobain). A photomultiplier tube (PMT, Hamamatsu, H7416) was used as a photodetector. All measurements were made at room temperature.

Neutron imaging using Ce doped LiCaAlF_6 scintillator and PSPMT which had 64 channel multi-anode were demonstrated. Obtained crystal was cut to the circular plate shape with the size of $\Phi 55$ mm and 2 mm thickness and optically coupled to PSPMT by silicone grease (OKEN 6262A). The sealed ^{252}Cf source (<1 MBq) was used with a thickness of 43 mm polyethylene for neutron thermalization. Alphabet-shape Cd pieces with a thickness of 4 mm were used as a mask of thermal neutron. Signal at each anode was fed into preamplifier, followed by the shaping amplifier with 5 μs integration and the personal computer.

2 Results and discussion

2.1 Pr:LuAG

Growth conditions such as pulling rate, rotation rate, and insulator design in the furnace were optimized for obtaining crack free 2-inch-diameter Pr:LuAG crystals with uniform light output and decay time in whole crystal. 2-inch-diameter 2.5%Pr:LuAG single crystals with a length of 110 mm was grown with a nominal composition of $(\text{Pr}_{0.075}\text{Lu}_{2.925})\text{Al}_5\text{O}_{12}$ (Fig. 1). The solidification yield of Pr:LuAG for raw materi-



Fig. 1 2-inch-diameter bulk Pr:LuAG single crystal grown by the Cz method

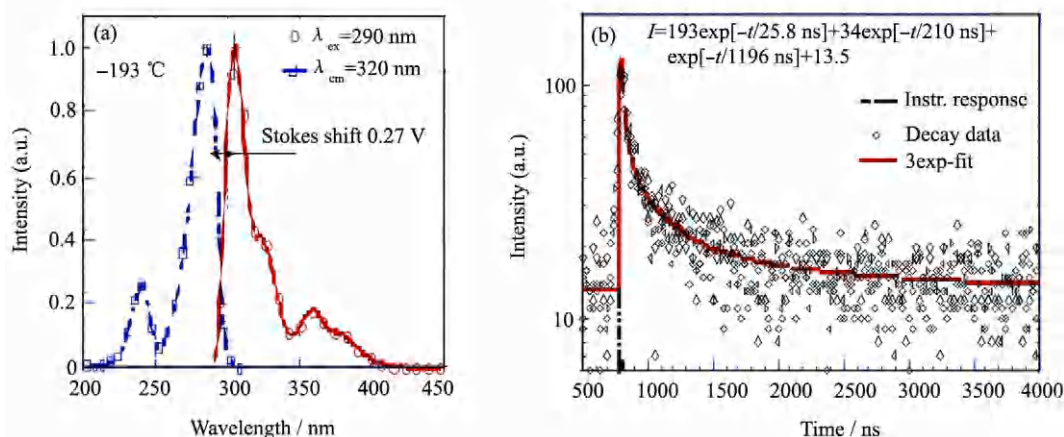


Fig. 2 Excitation and emission spectra of Pr:LuAG (a) and scintillation decay of Pr:LuAG (b)

als was about 40% of melt of raw material in the crucible.

Excitation and emission spectra of Pr:LuAG is shown in Fig. 2(a). Pr^{3+} $5d \rightarrow 4f$ transitions are dominating both spectra^[11] and the Stokes shift is found as 0.27 eV. Scintillation decay of Pr:LuAG dominated by 26 ns decay time is shown in (b), slower component with 210 and 1200 ns decay time are clearly visible as well.

The ^{137}Cs gamma-ray image using a test detector consisting of 20×20 pixelated array is shown in Fig. 3.

1.3.2 Ce:LiCAF and Eu:LiCAF Growth conditions such as pulling rate, rotation rate, and insulator design in the furnace were optimized for obtaining crack free 2-inch-diameter Ce and Eu doped LiCAF crystals with Colquiriite-type structure. As shown in Fig. 4, 2-inch-diameter 2.0%Ce:LiCAF and 2%Eu:LiCAF single crystals (nominal composition) with a length of 100 mm were grown by the Cz method. Both

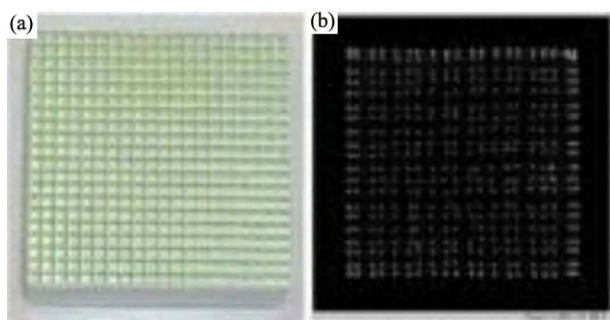


Fig. 3 20×20 Pr:LuAG pixelated array (a) and 2-dimensional image under ^{137}Cs irradiation (b)

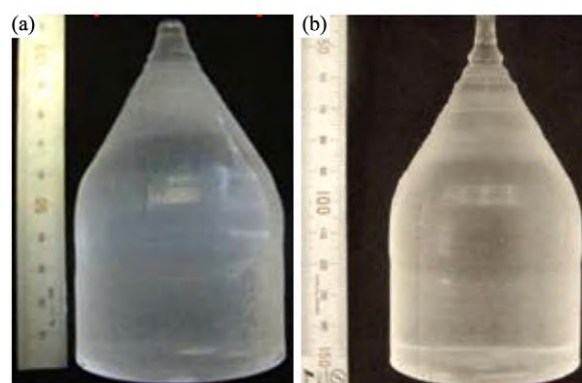


Fig. 4 2-inch-diameter bulk Ce:LiCAF (a) and Eu:LiCAF single crystal (b) grown by the Cz method

crystal were transparent and have no visible inclusions and cracks.

Transmission and emission spectra of Ce:LiCAF and Eu:LiCAF are shown in Fig. 5. Ce^{3+} and Eu^{2+} $5d-4f$ transitions are clearly observed. The observed characteristics are consistent with earlier reports at these materials studied under UV and gamma excitations^[17,18]. While the decay time of the Ce^{3+} center in LiCAF host is of about 20–35 ns^[17], and that of Eu^{2+} is longer of about 1–1.2 μs ^[16,18].

α/β ratio of Ce:LiCAF and Eu:LiCAF crystals are shown in Fig. 6. It was observed that the α/β ratios for Ce:LiCAF or Eu:LiCAF are different. This was found by the Soltan Institute for Nuclear Studies^[19].

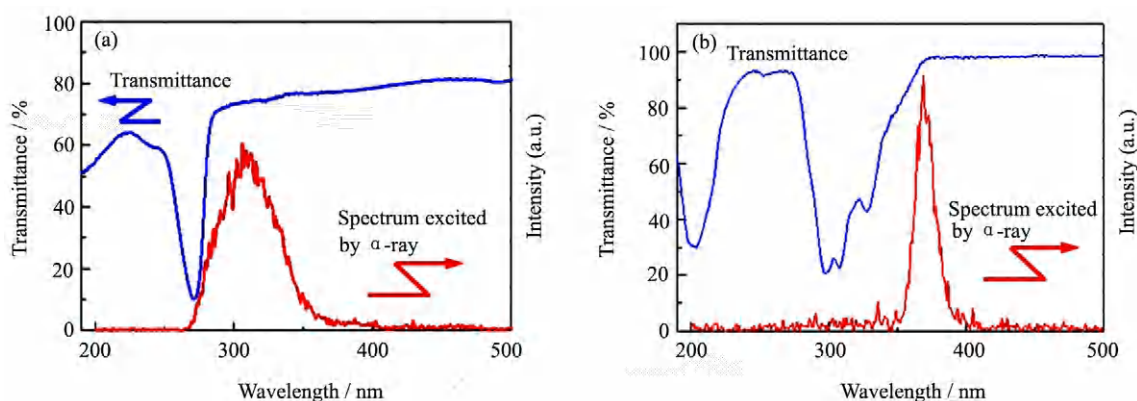
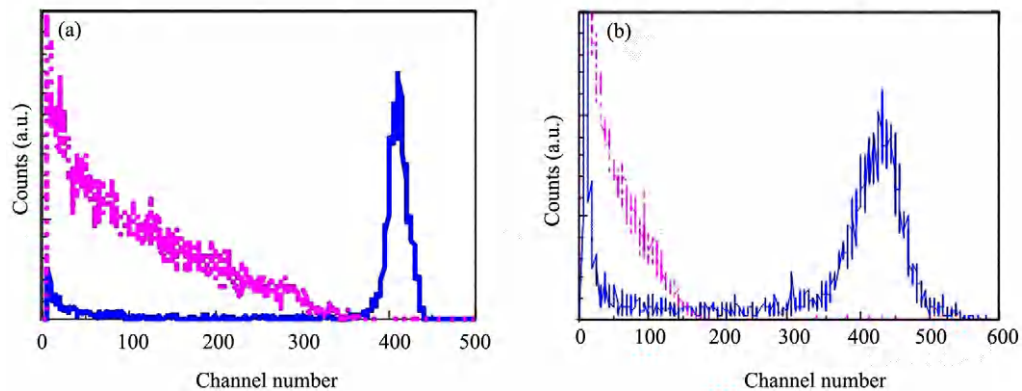
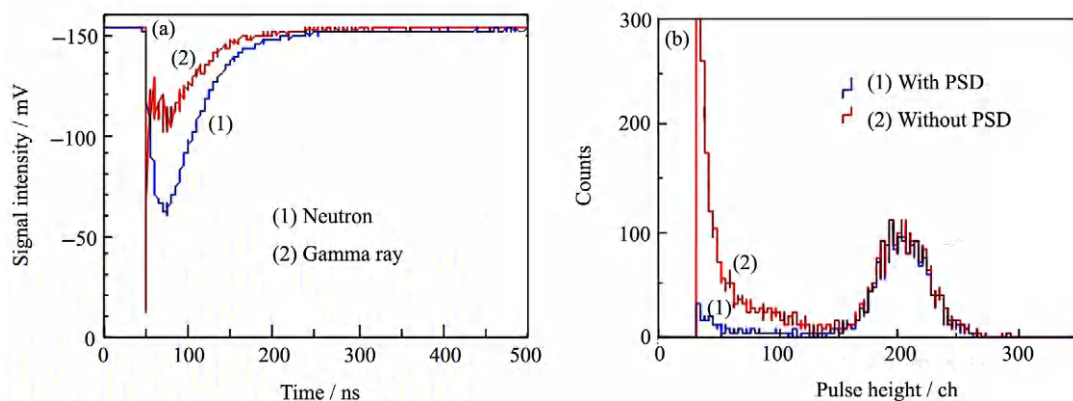
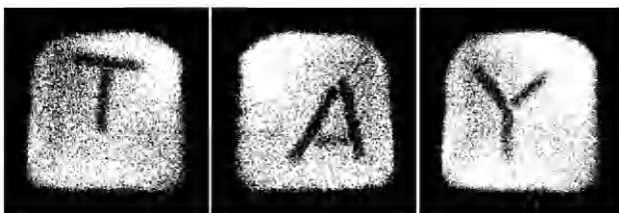


Fig. 5 Transmittance and emission spectra of Ce:LiCAF (a) and Eu:LiCAF (b)

Fig. 6 α/β ratio of Ce:LiCAF (a) and Eu:LiCAF crystals (b)Fig. 7 Different pulse shapes for neutron and γ -ray irradiation in Ce:LiCAF (a) and pulse height spectra of Ce:LiCAF with and without PSD (b)

Nagoya University group found that Ce:LiCAF has different pulse shapes for neutron and γ -ray irradiation, and pulse shape discrimination were carried out as shown in Fig. 7^[20,21]. This feature is of great advantage to better discriminate the inevitable gamma background in neutron fluxes^[21].

As shown in Fig. 8, two dimensional neutron imaging was obtained using LiCAF plate with PS-PMT. The parallel beam of thermal neutron source at JRR-3, MUSASI (JAEA) was used. The neutron flux was around 8×10^5 counts/s.

Fig. 8 Two dimensional neutron imaging using the Cd mask^[16]

3 Conclusions

Crystal growth, absorption, photo- and radioluminescence spectra, as well as the performance test of Pr^{3+} :LuAG gamma-ray scintillator and Ce^{3+} : LiCAF and Eu^{2+} : LiCAF neutron scintillator were surveyed. Due to the mastered growth technology of big enough single crystals and competitive combination of scintillation parameters, these materials have great practical prospectives. Pr^{3+} in LuAG provided fast $5d \rightarrow 4f$ transition providing noticeably shorter decay time with respect to that of Ce^{3+} . As LuAG showed the best performance, after the development of technology on

bulk crystal growth, basic scintillation properties and two-dimensional gamma-ray imaging, positron emission mammography (PEM) was conducted on Pr^{3+} :LuAG. As a candidate for alternatives of ^3He neutron detectors, Ce and Eu doped LiCAF appeared competitive candidates. For this purpose, the discrimination ability of gamma-ray from alpha-ray is important. Therefore, pulse shape discrimination was investigated and succeeded. Bulk crystal growth, basic scintillation properties and two-dimensional neutron imaging were demonstrated, as well.

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