Contents lists available at ScienceDirect







journal homepage: www.elsevier.com/locate/jlumin

Spectral characterization and energy levels of Cr³⁺:Sc₂(MoO₄)₃ crystal

Guojian Wang, Zhoubin Lin, Lizhen Zhang, Yisheng Huang, Guofu Wang*

State Key Laboratory of Structural Chemistry, Fujian Institute of Research on the Structure of Matter, Chinese Academy of Sciences, Fuzhou, Fujian 350002, China

ARTICLE INFO

ABSTRACT

gain medium.

Article history: Received 24 January 2009 Received in revised form 7 July 2009 Accepted 10 July 2009 Available online 18 July 2009

PACS: 42.70.Нј 78.20.-е

Keywords: Spectral properties Cr³⁺:Sc₂(MoO₄)₃ crystal

1. Introduction

Since tunable solid-state lasers have a wide field application in medicine, ultra short pulse generation, environment and communication [1,2], research on Cr^{3+} -doped tunable solid-state laser in the visible and near infrared spectrum has generated strong interest. The Cr^{3+} as a active ion tends to be incorporated into environments which are octahedrally coordinated by ligands. Since in an oxide compound with general formula $A_2(BO_4)_3$ (A = Al, Sc; B = W, Mo) there are AlO₆ and ScO₆ octahedron, $A_2(BO_4)_3$ (A = Al, Sc; B = W, Mo) compounds can be considered as the Cr^{3+} -doped host materials. The spectroscopic properties of Cr^{3+} :Al₂(WO₄)₃ and Cr^{3+} :Sc₂(WO₄)₃ crystals were investigated as a tunable solid-state laser material [3,4]. Sc₂(MOO₄)₃ crystal belongs to the orthorhombic symmetry with space group Pnca [5–7]. This paper reports the optical properties of Cr^{3+} :Sc₂(MOO₄)₃ crystals.

2. Experimental procedure

 $Cr^{3+}:Sc_2(MoO_4)_3$ crystals were grown from a flux of Li₂Mo₂O₇ by the top seeded solution growth (TSSG) method at a cooling rate 1-2 K/d and a rotating rate of 4.5 rpm. The Cr^{3+} ions concentration in $Cr^{3+}:Sc_2(MoO_4)_3$ crystal was determined to be 0.61 at% by ionic coupled plasma (ICP) spectrometry. A sample with dimension of $5 \times 4 \times 2$ mm³ was cut from as-grown crystal for spectral

* Corresponding author. E-mail address: wgf@ms.fjirsm.ac.cn (G. Wang).

0022-2313/\$ - see front matter © 2009 Elsevier B.V. All rights reserved. doi:10.1016/j.jlumin.2009.07.012

measurement. The absorption spectrum was recorded using a Perkin-Elmer UV-vis-NIR Spectrometer (Lambda-900) at room

Perkin-Elmer UV-vis-NIR Spectrometer (Lambda-900) at room temperature. The fluorescence spectrum and fluorescence lifetime were measured using an Edinburgh Instruments FLS920 spectrophotometer with a continuous Xe-flash lamp and R928-P detector at 300 and 10 K.

© 2009 Elsevier B.V. All rights reserved.

3. Experimental results and discussion

This paper reports the spectral properties and energy levels of $Cr^{3+}:Sc_2(MoO_4)_3$ crystal. The crystal field

strength Dq, Racah parameter B and C were calculated to be 1408 cm^{-1} , 608 cm^{-1} and 3054 cm^{-1} ,

respectively. The absorption cross sections σ_{α} of ${}^{4}A_{2} \rightarrow {}^{4}T_{1}$ and ${}^{4}A_{2} \rightarrow {}^{4}T_{2}$ transitions were 3.74×10^{-19}

cm² at 499 nm and 3.21×10^{-19} cm² at 710 nm, respectively. The emission cross section σ_e was

 375×10^{-20} cm² at 880 nm. Cr³⁺:Sc₂(MoO₄)₃ crystal has a broad emission band with a broad FWHM of

176 nm (2179 cm⁻¹). Therefore, $Cr^{3+}:Sc_2(MoO_4)_3$ crystal may be regarded as a potential tunable laser

Fig. 1 shows the absorption spectrum of $Cr^{3+}:Sc_2(MoO_4)_3$ crystals. Two broad absorptions with peak at $\lambda = 499$ and 710 nm are due to the ${}^{4}A_2 \rightarrow {}^{4}T_1$ and ${}^{4}A_2 \rightarrow {}^{4}T_2$ transitions of Cr^{3+} ions, respectively. The two structures on either side of the ${}^{4}T_2$ absorption peak are due to absorption into the ${}^{2}T_1$ and ${}^{2}E$ states, which show up as Fano antiresonance structures [8,9]. We take the dips in the absorptions. The absorption cross sections σ_{α} of ${}^{4}A_2 \rightarrow {}^{4}T_1$ and ${}^{4}A_2 \rightarrow {}^{4}T_2$ transitions were estimated to be $3.74 \times 10^{-19} cm^2$ at 499 nm and $3.21 \times 10^{-19} cm^2$ at 710 nm, respectively.

The Cr³⁺ ions substitute at Sc³⁺ site in Sc₂(MoO₄)₃, where the symmetry is approximately octahedral. The peak energy of the ${}^{4}A_{2} \rightarrow {}^{4}T_{2}$ band measures 10*Dq* [10], thus the *Dq* was measured to be 1408 cm⁻¹ from the spectrum of Fig. 1. The energy at the peak of the ${}^{4}A_{2} \rightarrow {}^{4}T_{1}$ band depends on both *Dq* and *B* [11]. The *B* can be calculated by the following equation [10]:

$$\frac{B}{Dq} = \frac{(\Delta/Dq)^2 - 10(\Delta/Dq)}{15(\Delta/Dq - 8)}$$
(1)



Fig. 1. Absorption spectrum of $Cr^{3+}:Sc_2(MoO_4)_3$ crystal at 300 K.

Table 1

Comparison of crystal field parameters for $Cr^{3\ast}:Sc_2(MoO_4)_3$ with other $Cr^{3\ast}\text{-doped crystals.}$

Material	Dq (cm ⁻¹)	<i>B</i> (cm ⁻¹)	Dq/B	$C(cm^{-1})$	C/B	
Al ₂ O ₃ [13]	1664	640	2.6	3300	5.2	
YAB [14]	1680	672	2.5	3218	4.8	
GAB [15]	1695	673	2.52	3380	5.0	
YSB [15]	1539	644	2.39	-	-	
GSB [15]	1563	638	2.45	-	-	
LSB [16]	1529	675	2.27	3448	5.1	
GSGG [17]	1563	638	2.45	-	-	
$Sc_2(WO_4)_3$ [4]	1449	630	2.3	-	-	
$Al_2(WO_4)_3$ [4]	1495	650	2.3	-	_	
$Sc_2(MoO_4)_3$	1408	608	2.32	3054	5.0	

where Δ is the difference in energy at the peaks of the two ${}^{4}A_{2} \rightarrow {}^{4}T_{1}$ and ${}^{4}A_{2} \rightarrow {}^{4}T_{2}$ bands, which is 5955 cm⁻¹. Substituting the measured values of Dq and Δ into Eq. (1), the value of B was calculated to be 608 cm⁻¹, from which the value of Dq/B was calculated to be 2.32. To calculate C requires a knowledge of the energy of the ${}^{2}E$ level, $E({}^{2}E)$. C is obtained from the formula [10]:

$$E(^{2}E) \simeq 3.05C + 7.90B - 1.80B^{2}/Dq$$
⁽²⁾

for the range of values 1.5 < Dq/B < 3.5 and 3 < C/B < 5, where $E(^{2}E)$ is the dip energy 13643 cm⁻¹ (733 nm) of broad band $^{4}A_{2} \rightarrow ^{4}T_{2}$ transitions of Cr^{3+} ions in $Cr^{3+}:Sc_{2}(MOO_{4})_{3}$ crystal[12]. Using obtained values of *B* and *Dq*, the *C* was calculated to be 3054 cm⁻¹. The values of *Dq*, *B* and *C* for $Cr^{3+}:Sc_{2}(MOO_{4})_{3}$ crystal are compared with other laser gain media in Table 1.

Fig. 2 shows the fluorescence spectrum of $Cr^{3+}:Sc_2(MOQ_4)_3$ crystals at 300 and 10 K. The dominant feature of fluorescence spectrum of $Cr^{3+}:Sc_2(MOQ_4)_3$ crystal is a broad band with a peak at 880 nm and a full width at half maximum (FWHM) of 176 nm (2179 cm⁻¹) corresponding to ${}^{4}T_2 \rightarrow {}^{4}A_2$ transition at 300 K. On cooling to 10 K, the fluorescence spectrum is still broad emission, showing that the ${}^{4}T_2$ zero-vibrational level is below the ${}^{2}E$ level. The weak field site in $Cr^{3+}:Sc_2(MOQ_4)_3$ crystal gives rise the Cr^{3+} fluorescence in the ${}^{4}T_2 \rightarrow {}^{4}A_2$ transition.

As well known, in the Cr^{3+} -doped tunable laser crystals the fluorescence line shape as a function of temperature is determined by the energy splitting $\Delta E = E({}^{4}T_{2})-E({}^{2}E)$, where $E({}^{4}T_{2})$ is the energy of the zero-vibrational ${}^{4}T_{2}$ state[17]. When the value of ΔE is large positive, the ${}^{4}T_{2}$ level is above ${}^{2}E$ level, the R-line and



Fig. 2. Photoluminescence spectrum of $Cr^{3+}:Sc_2(MoO_4)_3$ measured excitation with 710 nm radiation at 300 and 10 K.

its vibronic sideband is only observed. For example, the laser action of ruby $(Cr^{3+}:Al_2O_3)$ is driven by the ${}^2E \rightarrow {}^4A_2$ R-line emission; for large negative value of ΔE the ${}^{4}T_{2}$ level is below ${}^{2}E$ level, the broad emission band of ${}^{4}T_{2} \rightarrow {}^{4}A_{2}$ transition is only observed as observed for Cr:LiCAF and Cr:LiSAF crystals; when the value of ΔE is close to $\Delta E \approx 0$, the observed emission band shape is an intimate mélange of R-line and the broadband process. For $Cr^{3+}:Sc_2(MoO_4)_3$ crystal, the peak of the ${}^4A_2 \rightarrow {}^4T_2$ absorption corresponds to a transition into a higher vibrational level ⁴T₂ state, and does not give the zero-phonon-transition level. Very approximately, one can predict that the position of the ${}^{4}A_{2} \rightarrow {}^{4}T_{2}$ zero-phonon transition is midway between the absorption and emission peaks, approximately 12724 cm^{-1} above the ${}^{4}\text{A}_{2}$ level. Then, $\Delta E = -918 \text{ cm}^{-1}$. Therefore, the ${}^{4}\text{T}_{2}$ level is below the ²E level, and a broad emission should be observed at all temperatures.

The emission cross section $\sigma_{\rm e}$ was calculated using the formula [11]

$$\sigma_{\rm e} = \frac{\lambda^2}{4\pi^2 \tau_{\rm f} n^2 \Delta \nu} \tag{3}$$

where λ is the wavelength of emission peak, *n* the refractive index which was estimated to be 2.0 by Abbe retractometer at 589 nm wavelength, Δv the frequency at FWHM ($\Delta v = 6.54 \times 10^{13} \text{ s}^{-1}$) and $\tau_{\rm f}$ the fluorescence lifetime. The fluorescence lifetime was measured to be 0.2 µs at 300 K. Thus, the emission cross section $\sigma_{\rm e}$ was calculated to be 375 × 10⁻²⁰ cm² at 880 nm at 300 K.

4. Conclusion

The spectral characterization of $Cr^{3+}:Sc_2(MoO_4)_3$ crystal was investigated. $Cr^{3+}:Sc_2(MoO_4)_3$ crystal exhibits a broad emission band. The emission band is extended from 750 to 1200 nm with a peak at 880 nm and an FWHM of 176 nm at 300 K. Of course, another major interest is the broad absorption bands of $Cr^{3+}:Sc_2(MoO_4)_3$ crystal, one of which is near $\lambda = 710$ nm due to $^{4}A_2 \rightarrow ^{4}T_2$ transition, which has the potential for efficient pumping using visible-range diode laser. In comparison with the other Cr^{3+} doped materials (see Table 2), $Cr^{3+}:Sc_2(MoO_4)_3$ crystal has a broad emission band, larger absorption and emission cross sections. $Cr^{3+}:Sc_2(MoO_4)_3$ crystal has a very short fluorescence lifetime, however, it can be used as a short pulse tunable laser gain medium. Therefore, $Cr^{3+}:Sc_2(MoO_4)_3$ crystal may be regarded as a potential tunable laser gain medium.

Table 2

Spectral parameters of Cr³⁺:Sc₃(MoO₄)₃ crystal and other Cr³⁺-doped materials.

Material	${}^{4}A_{2} \rightarrow {}^{4}T_{1}$		${}^{4}A_{2} \rightarrow {}^{4}T_{2}$	${}^{4}A_{2} \rightarrow {}^{4}T_{2}$		${}^{4}T_{2} \rightarrow {}^{4}A_{2}$			
	λ (nm)	σ_{α} (10 ⁻²⁰ cm ²)	λ (nm)	σ_{α} (10 ⁻²⁰ cm ²)	λ (nm)	FWHM	$\sigma_{\rm e}~(10^{-20}{\rm cm}^2)$	$\tau_{f}(\mu s)$	
BeAl ₂ O ₄	420	10.0	600	20.0	750	_	0.5	260	[12]
K ₂ NaScF ₆	430	1.4	630	0.7	760	-	1.3	285	[12]
GSGG	488	5.1	647.1	3.3	777	_	0.75	114	[18]
LiCaAlF ₆	425.5	-	625		780	$2000 \mathrm{cm}^{-1}$	1.23	175	[19]
$LaSc_3(BO_3)_4$								17	[11]
Ellx	457	1.18	654	1.68	948	200 nm	6.13		
Elly	456	1.72	655	1.01	948	210 nm	5.83		
Ellz	458	1.32	655	0.81	948	280 nm	4.33		
$KAl(MoO_4)_2$								33	[20]
σ -Polarization	480	8.44	669	3.72	823	146 nm	2.74		
π -Polarization	481	5.03	668	2.25	823	135 nm	2.93		
$Cr^{3+}:Sc_2(MoO_4)_3$	499	37.4	710	32.1	880	176 nm (2179 cm ⁻¹)	375	0.2	This work

Acknowledgments

This work is supported by the National Natural Science Foundation of China (No. 60778035) Key Project of Science and Technology of Fujian Province (2001F004) and the Young Scientists Innovation Foundation of Fujian Province (2008F3113), respectively.

References

- [1] S. Kück, Appl. Phys. B 72 (2001) 515.
- [2] T.A. Samtleben, J. Hulliger, Opt. Laser Eng. 43 (2005) 251.
- [3] K. Petermann, G. Huber, J. Lumin. 31/32 (1984) 71.
- [4] K. Petermann, P. Mitzscherlich, IEEE J. Quantum Electron. QE-23 (1987) 1122.
- [5] A.W. Biedl, Acta Crystallogr. 19 (1965) 1059.
- [6] V.A. Efremov, B.I. Lazoryak, V.K. Trunov, Kristallografiya 26 (1981) 72. [7] J.S. Evans, T.A. Mary, Int. J. Inorg. Mater. 2 (2000) 143.
- [8] U. Fano, Phys. Rev. 124 (1961) 1866.

- [9] U. Fano, J.W. Cooper, Phys. Rev. 137 (1965) A1364.
- [10] B. Henderson, G.F. Imbush, Optical Spectroscopy of Inorganic Crystals, Oxford University Press, Oxford, 1989.
- [11] X.F. Long, Z.B. Lin, Z.S. Hu, G.F. Wang, Chem. Phys. Lett. 392 (2004) 192.
- [12] P.T. Kenyon, L. Andrews, B. McCollum, A. Lempicki, IEEE J. Quantum Electron. QE18 (1982) 1189.
- [13] G.F. Imbush, R. Kogelman, Laser spectroscopy of solids, in: W.A. Yen, P.M. Selzer (Eds.), Topics Applied Physics, Springer, Berlin, 1981.
- [14] G. Wang, H.G. Gallagher, T.P.J. Han, B. Henderson, J. Cryst. Growth 153 (1995) 169.
- [15] G. Wang, H.G. Gallagher, T.P.J. Han, B. Henderson, J. Cryst. Growth 183 (1996) 272.
- [16] X.F. Long, Z.B. Lin, Z.S. Hu, G.F. Wang, J. Alloys Compd. 347 (2002) 52.
- [17] B. Struve, G. Huber, Appl. Phys. B 36 (1985) 195.
 [18] B. Struve, G. Huber, V.V. Laptev, I.A. Shcherbakov, E.V. Zharikov, Appl. Phys. B 30 (1983) 117.
- [19] S.A. Payne, L.L. Chase, H.W. Newkirk, L.K. Smith, W.F. Krupke, IEEE J. Quantum Electron QE-24 (1998) 2243.
- [20] G.J. Wang, X.F. Long, L.Z. Zhang, G.F. Wang, S. Polosan, T. Tsuboi, J. Lumin. 128 (2008) 1556.